**Vol. 7, No. 1, 2024** 

**Yu. V. Sukhatskyi<sup>1</sup> , M. A. Sozanskyi<sup>2</sup> , M. V. Shepida<sup>1</sup> , Z. O. Znak<sup>1</sup> , S. V. Khomyak<sup>3</sup>**

Lviv Polytechnic National University,

<sup>1</sup> Department of Chemistry and Technology of Inorganic Substances,

 $2^{2}$  Department of Physical, Analytical and General Chemistry,

<sup>3</sup> Department of Technology of Biologically Active Substances, Pharmacy and Biotechnology

yurii.v.sukhatskyi@lpnu.ua

# **SYNTHESIS OF SPINEL MgMn2O<sup>4</sup> NANOPARTICLES BY THE CO-PRECIPITATION METHOD IN AN ULTRASONIC FIELD**

https://doi.org/10.23939/ctas2024.01.052

**Nanoparticles of spinel MgMn2O<sup>4</sup> were synthesized using the co-precipitation method in an ultrasonic field. It was established that at a calcination temperature of 200 °C, all peaks on the diffractogram of the synthesized material corresponded to spinel MgMn2O<sup>4</sup> with a cubic lattice, pronounced crystallinity, and the absence of other phases. As the calcination temperature**  increased, the formation of new phases – Mn oxides (respectively,  $Mn_5O_8$  and  $Mn_2O_3$ ) – was **recorded. The average size of MgMn2O<sup>4</sup> particles was calculated from the diffraction peaks using the Debye-Scherrer equation and equated to 24.4 nm at a calcination temperature of 200 °C. An increase in the specific power of the ultrasonic processing of the reaction medium revealed a natural increase in the proportion of the amorphous phase and a decrease in the average size of MgMn2O<sup>4</sup> particles.**

**Key words: spinel; nanoparticles; co-precipitation method; ultrasonic field; Debye-Scherrer equation; average crystallite size.**

## **Introduction**

Spinels are materials with the general formula AB<sub>2</sub>O<sub>4</sub>, where A =  $Zn^{2+}$ , Mg<sup>2+</sup>, Cu<sup>2+</sup>, Co<sup>2+</sup> and  $B = Co^{3+}$ ,  $Mn^{3+}$ ,  $Al^{3+}$ ,  $Fe^{3+}$ , or other metals [1]. Nanoscale spinels have many practical applications. Thus, nanocrystalline ferrites with a spinel structure are used as catalysts in gas sensors, magnetic information carriers, energy storage devices (supercapacitors), magnetic hyperthermia, and controlled drug delivery systems [2]. Possible areas of application of  $MnFe<sub>2</sub>O<sub>4</sub>$  nanoparticles and other Mn-containing spinels include the following: production of lithiumion batteries, heavy metal adsorption processes, high-speed generation of hydrogen due to thermochemical decomposition of water [3]; heterogeneous activation of oxidants (hydrogen peroxide, peroxymonosulfates) in the processes of degradation of dyes [4–6], benzyl alcohol [7], bisphenol A [8, 9], antibiotics (erythromycin, tetracycline, amikacin, ofloxacin) [10, 11]; inactivation of microorganisms – gram-positive bacteria (*Staphylococcus aureus*, *Streptococcus pneumonia*), gram-negative bacteria

(*Pseudomonas aeruginosa*, *Salmonella paratyphi*), fungi (*Candida albicans*) [7].

The method and conditions of synthesis determine the size, morphology, purity, and crystallinity of MnFe<sub>2</sub>O<sub>4</sub> spinel nanoparticles, their stability and electromagnetic characteristics [3, 12]. The most common methods of  $MnFe<sub>2</sub>O<sub>4</sub>$  synthesis include [3, 12, 13] the co-precipitation method, solvothermal, hydrothermal, sol-gel methods, thermal decomposition, microemulsion, polyol, template synthesis, the solidphase reaction method, etc. The advantages of the co-precipitation method include its simplicity, low cost, low temperature and short duration of synthesis, as well as high ability to obtain homogeneous particles of metal multioxides [12, 13].

Catalysts of the spinel type, which contain transition elements (metals) with a 3d-sublevel and a second metal, are more active and stable in advanced oxidation processes due to the synergistic effect of the two oxides, compared to monooxides. Thus, spinel  $MgMn_2O_4$  was synthesized using the coprecipitation method, which demonstrated high

catalytic activity during the activation of sodium periodate in the process of oxidative degradation of bisphenol A [1]. The degree of degradation of bisphenol A was equal to 96.17 % due to the use of the  $MgMn_2O_4/NaIO_4$  system for 60 minutes. The conditions of oxidative degradation were as follows: reaction volume  $-100$  mL; pH of the environment  $-$ 7.00; temperature – 20 °C; concentration of bisphenol  $A - 10^{-5}$  mol/L; concentration of NaIO<sub>4</sub> – 10<sup>-3</sup> mol/L; concentration of  $MgMn_2O_4$  is 0.1 g/L. Magnesium oxide, which is part of the spinel, is widespread in nature and has low toxicity in water bodies. Therefore,  $MgMn_2O_4$  is an environmentally friendly catalyst that can be used to activate oxidizing agents.

Ultrasound is a method of targeted influence on the morphology and size of spinel particles during their synthesis. Therefore, **the aim of the work** was to study the structural and phase characteristics of  $MgMn_2O_4$  spinel particles synthesized by the coprecipitation method in an ultrasonic field.

#### **Materials and research methods**

The synthesis of  $MgMn_2O_4$  spinel nanoparticles was carried out using the method of coprecipitation in an ultrasonic (US) field. Magnesium nitrate hexahydrate  $(Mg(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O)$  and manganese (II) nitrate hexahydrate  $(Mn(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O)$  were used as precursors for the synthesis. First,  $Mg(NO_3)$ <sup>2</sup>·6H<sub>2</sub>O and  $Mn(NO_3)$ <sup>2</sup>·6H<sub>2</sub>O were dissolved in 300 mL of distilled water in a molar ratio of 1:2 with constant stirring. Then the solution was heated to 75 °C in a water bath, the ultrasonic oscillation generator was turned on, and a precipitation agent (aqueous solution of sodium hydroxide with a concentration of 2 mol/L) was added dropwise with constant stirring until the pH value of the reaction medium equated to 9.00. After that, the effect of the ultrasound field on the reaction medium and the addition of drops of sodium hydroxide was stopped. The resulting suspension, which contained light brown particles, was stirred for 2 hours at a temperature of 75 °C to ensure complete crystallization and growth of nanoparticles. To obtain a pure product that does not contain unwanted impurities, the synthesized materials were washed three times with distilled water and three times with ethanol until the pH value equated to 7.00, then separated from the liquid phase by centrifugation (duration  $-5$  min;

frequency – 5000 rpm), and dried at a temperature of 60 °C for 12 hours. The obtained nanoparticles were annealed in an air atmosphere at different temperatures (respectively, 200, 400, 600, and 800 °C) for 5 h. with a heating rate of 5 °C/min.

A magnetostriction-type emitter (Bandelin Sonopuls HD 2200.2, Germany) was used as a generator of US-oscillations (frequency – 20 kHz). The specific power of ultrasonic processing of the reaction medium varied from 200 to 600 W/dm<sup>3</sup>. The pH values were monitored with an ADWA AD1200 ATC pH meter with a combined glass electrode and thermocompensator.

The phase composition, average crystallite size, and crystal lattice parameters of the samples under different synthesis conditions were studied by X-ray diffraction (XRD) using an AERIS Research diffractometer (Malvern PANalytical, Great Britain) with Cu Kα radiation ( $\lambda = 1.5406$  Å). The average crystallite size (*D*, nm) of the synthesized materials was estimated using the Debye-Scherrer equation [13, 14]:

$$
D = \frac{k\lambda}{\beta \cos \theta},\tag{1}
$$

where  $k$  is Scherrer's constant, which depends on the Miller index, the index of the reflection plane and the shape of the crystal (usually,  $k = 0.9$ );  $\lambda$  is the wavelength of X-radiation ( $\lambda = 1.5406$  Å); β is the full width at half maximum of the reflection peaks, rad.;  $\theta$  – diffraction angle (Bragg angle), rad.

Checking the mass and atomic fractions of Mg, Mn, and O in the obtained samples was carried out by the method of energy dispersive X-ray analysis (EDAX) using the X-ray fluorescence analyzer ElvaX Pro (Elvatech, Ukraine).

The scanning electron microscopy (SEM) method was used to study the morphology of the synthesized samples (Tescan Vega 3 LMU electron microscope (TESCAN Brno s.r.o., Czech Republic), equipped with SE- (surface topographic contrast) and BSE-detectors (phase electron contrast)).

Functional groups of synthesized materials were analyzed by Fourier transform infrared spectroscopy (FTIR) (Spectrum Two spectrometer (Perkin Elmer, USA)).

## **Results and discussion**

The formation of  $MgMn_2O_4$  nanoparticles occurs as a result of the interaction of metal nitrates

and the precipitating agent (NaOH) according to reactions (2)–(6) [12, 15]:

$$
Mg(NO3)2 + Mn(NO3)2 + 4NaOH \rightarrow
$$
  
\n
$$
\rightarrow Mg(OH)2 + Mn(OH)2 + 4NaNO3,
$$
 (2)

$$
2\text{Mn(OH)}_2 + 2\text{OH}^- \rightarrow 2\text{Mn(OH)}_3 + 2\text{e}^-, \qquad (3)
$$

$$
Mg(OH)_2 \leftrightarrow Mg^{2+} + 2OH^-, \tag{4}
$$

$$
Mn(OH)_3 \leftrightarrow Mn^{3+} + 3OH^-, \tag{5}
$$

$$
Mg^{2+} + 2Mn^{3+} + 8OH^- \rightarrow MgMn_2O_4 + 4H_2O. \tag{6}
$$

The crystal structure and phase purity of the synthesized materials at different calcination temperatures was evaluated using the X-ray diffraction method (Fig. 1).

All diffraction peaks at a calcination temperature of 200 °C corresponded to spinel  $MgMn_2O_4$  with a cubic lattice, pronounced crystallinity, and the absence of other phases (Fig. 1). As the calcination temperature increased to 400 °C, the appearance of  $Mn_5O_8$  peaks and a significant increase in the proportion of the amorphous phase were observed in the diffractogram. A further increase in the calcination temperature to 600 °C led

to the decomposition of  $Mn_5O_8$  with the formation of  $Mn<sub>2</sub>O<sub>3</sub>$ . However, it should be noted that at calcination temperatures of 400 and 600 °C, the molar fraction of spinel  $MgMn_2O_4$  (61.62 and 91.30 %, respectively) significantly exceeded the molar fraction of other phases –  $Mn_5O_8$  (38.38 %) and  $Mn_2O_3$  (8.70 %)) (Table 1).



*Fig. 1. Diffractograms of materials synthesized by the co-precipitation method in the US field at different calcination temperatures (specific power of ultrasound processing of the reaction medium is 200 W/L)*

*Table 1*

Calcination temperature, $\rm ^{\circ}C$	Phase	Mole fraction of the phase, $\frac{0}{0}$	Lattice parameters,	β, rad.	$\cos \theta$	D, nm
200	$MgMn_2O_4$	100.00	$a = 5.762$ (1); $c = 9.353$ (1)	0.005974	0.95075	24.4
400	$MgMn_2O_4$	61.62	$a = 5.764$ (1); $c = 9.340$ (1)	0.010645	0.95078	13.7
	$Mn_5O_8$	38.38	$a = 10.368(3);$ $b = 5.738(2) - angle = 109.73(2)$ $deg.$ ; $c = 4.876(1)$	0.012504	0.96887	11.4
600	$MgMn_2O_4$	91.30	$a = 5.752$ (1); $c = 9.407$ (1)	0.005589	0.95065	26.1
	$Mn_2O_3$	8.70	$a = 9.400(1)$	0.003632	0.95891	39.8
800	$MgMn_2O_4$	54.34	$a = 5.757(1); c = 9.416(1)$	0.005964	0.95073	24.5
	$Mn_2O_3$	45.66	$a = 9.399(1)$	0.003508	0.95889	41.2

**Structural and phase characteristics of materials synthesized using the co-precipitation method in the ultrasound field (specific ultrasound power – 200 W/L)** 

At a calcination temperature of 800 °C, the ratio of spinel mole fractions  $MgMn_2O_4$  (54.34 %) and  $Mn_2O_3$  (45.66 %) was significantly lower than at lower calcination temperatures.

Therefore, with an increase in the calcination temperature, the intensity of spinel decomposition into individual oxides increased. This is also confirmed by the EDAX results. Thus, even at a calcination temperature of 400  $^{\circ}$ C, a violation of the stoichiometric composition of  $MgMn_2O_4$  spinel was recorded, which consisted of an increase in the mass or atomic fractions of Mn and O due to the formation of  $Mn_5O_8$  (Fig. 2; Table 2). Therefore, 200 °C was chosen as the rational value of the calcination temperature at which a single-phase system  $(MgMn<sub>2</sub>O<sub>4</sub> spinel)$  was obtained.



*Fig. 2. EDAX spectrum of the material synthesized using the co-precipitation method in* the US field (calcination temperature  $-400$  °C)

*Table 2*

**Comparison of the elemental composition of the synthesized sample (based on the results of the EDAX method) with the stoichiometric composition of spinel MgMn2O<sup>4</sup>**

Element	Material synthesized by the co-precipitation method in US field		$MgMn2O4$ spinel of stoichiometric composition		
	composition, % wt.	composition, % at.	composition, % wt.	composition, % at.	
Mg				14.3	
Mn	59.3	30.2	55.6	28.6	
	38.2	67 0	29 Z		

The main structural and phase characteristics (qualitative and quantitative composition of phases, crystal lattice parameters, average size of crystallites) of the samples synthesized by coprecipitation method in the ultrasound field are given in Table 1. The average size of  $MgMn_2O_4$  particles obtained at a calcination temperature of 200 °C was 24.4 nm. At higher calcination temperatures, the average size of  $MgMn<sub>2</sub>O<sub>4</sub>$  particles, calculated from diffraction peaks using the Debye – Scherrer equation, was in the range of 13.7–26.1 nm.

Based on the analysis of SEM images of  $MgMn_2O_4$  particles (Fig. 3) and the EDAX maps of the spinel (Fig. 4), it was established that  $MgMn_2O_4$ nanoparticles are evenly distributed, uniform in size and morphology (their shape is spherical), but agglomerated. As the calcination temperature of the samples increased, the tendency to agglomeration of nanosized  $MgMn<sub>2</sub>O<sub>4</sub>$  particles also increased. This can be explained by the strengthening of the interaction between particles with magnetic properties, in particular between particles of Mn oxides [14].

The FTIR spectrum of spinel  $MgMn_2O_4$ (Fig. 5) was recorded in the range of wave numbers  $400-4000$  cm<sup>-1</sup>. The analysis of this spectrum shows the presence of peaks at 473 and  $600 \text{ cm}^{-1}$ , indicating the valence vibrations of Mg–O and Mn–O, respectively. Peaks at 1635 and 3348  $cm^{-1}$  correspond to valence vibrations of hydroxyl groups of adsorbed water and hydroxyl groups on the surface of  $MgMn<sub>2</sub>O<sub>4</sub>$ .

The influence of the specific power of the ultrasonic processing of the reaction medium on the size of  $MgMn_2O_4$  spinel particles was determined based on the results of calculations based on diffractograms (Fig. 6).



*Fig. 4. EDAX maps of MgMn2O<sup>4</sup> spinel (calcination temperature – 200 °С)*



*Fig. 5. FTIR spectrum of*  $MgMn_2O_4$  *spinel (calcination temperature – 200 °C)* 



*Fig. 6. Diffractograms of MgMn2O<sup>4</sup> spinel synthesized by co-precipitation method at a different specific power of US processing of the reaction medium*

*Table 3*

**The influence of the specific power of the ultrasonic processing of the reaction medium on the average size of MgMn2O<sup>4</sup> spinel particles synthesized by the co-precipitation method in the ultrasonic field** 



As the specific power of ultrasonic processing of the reaction medium increased, the proportion of the amorphous phase increased (Fig. 6), and the average size of  $MgMn_2O_4$  particles naturally decreased (Table 3). This was caused by the intensification of the dispersive effect of the ultrasound with an increase in its power.

### **Conclusions**

 $MgMn<sub>2</sub>O<sub>4</sub>$  spinel particles were synthesized by co-precipitation method in the ultrasonic field. X-ray diffraction, energy dispersive X-ray analysis, scanning electron microscopy, and Fourier transform infrared spectroscopy were used to characterize the synthesized particles. It was established that at a calcination temperature of 200 °C, all peaks on the diffractogram of the synthesized material corresponded to spinel  $MgMn_2O_4$  with a cubic lattice, pronounced crystallinity, and the absence of other phases. With an increase in the calcination temperature, the formation of new phases – Mn oxides (respectively,  $Mn_5O_8$  and  $Mn_2O_3$ ) – was recorded, which is due to the decomposition of the spinel. The average size of  $MgMn_2O_4$  particles obtained at a calcination temperature of 200 °C was 24.4 nm.

Based on the analysis of SEM images of  $MgMn_2O_4$  particles and EDAX maps of the spinel, it was found that  $MgMn_2O_4$  nanoparticles are evenly distributed, uniform in size and morphology (their shape is spherical), but agglomerated. As the calcination temperature of the samples increased, the tendency to agglomeration of nanosized  $MgMn_2O_4$ particles increased. This can be explained by the strengthening of the interaction between particles with magnetic properties, in particular between particles of Mn oxides.

An increase in the specific power of ultrasonic processing of the reaction medium from 200 to 600 W/L led to an increase in the proportion of the amorphous phase and a decrease in the average size of  $MgMn_2O_4$  particles (calcination temperature – 200 °C) from 24.4 to 15.3 nm.

*Authors acknowledge the funding of Ministry of Education and Science of Ukraine for scientific research project of young scientists "Advanced oxidation processes, including nanocatalytic, based on cavitation technologies for purification of aqueous media from resistant N-substituted organic compounds" (state registration number 0122U000790).* 

*This study was conducted using devices of the Centre of Collective Use of Scientific Equipment: "Laboratory of Perspective Technologies of Creation and Physico-Chemical Analyses of New Substances and Functional Materials" at Lviv Polytechnic National University (https://lpnu.ua/ckkno).* 

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Ю. В. Сухацький<sup>1</sup>, М. А. Созанський<sup>2</sup>, М. В. Шепіда<sup>1</sup>, З. О. Знак<sup>1</sup>, С. В. Хом'як<sup>3</sup>

Національний університет "Львівська політехніка",

 $^{-1}$ кафедра хімії і технології неорганічних речовин,

 $2 \text{ кафедра фізичної, аналітичної та загальної хімії, }$ 

<sup>3</sup> кафедра технології біологічно активних сполук, фармації та біотехнології

#### **СИНТЕЗ НАНОЧАСТИНОК ШПІНЕЛІ MgMn2O<sup>4</sup> МЕТОДОМ СПІВОСАДЖЕННЯ В УЛЬТРАЗВУКОВОМУ ПОЛІ**

**Методом співосадження в ультразвуковому полі синтезовано наночастинки шпінелі MgMn2O4. Встановлено, що за температури кальцинації 200 °С усі піки на дифрактограмі синтезованого матеріалу відповідали шпінелі MgMn2O<sup>4</sup> з кубічною решіткою, вираженою кристалічністю та відсутністю інших фаз. З підвищенням температури кальцинації зафіксовано утворення нових фаз –**  оксидів Мп (відповідно, Мп<sub>5</sub>O<sub>8</sub> і Мп<sub>2</sub>O<sub>3</sub>). За дифракційними піками з використанням рівняння **Дебая – Шеррера розраховано середній розмір частинок MgMn2O4, який за температури кальцинації 200 °С дорівнював 24,4 нм. Виявлено закономірне збільшення частки аморфної фази і зменшення середнього розміру частинок MgMn2O<sup>4</sup> зі збільшенням питомої потужності ультразвукового оброблення реакційного середовища.** 

**Ключові слова: шпінель; наночастинки; метод співосадження; ультразвукове поле; рівняння Дебая – Шеррера; середній розмір кристаліту.**